Brazeability of the 6061-T6 Aluminum Alloy with Al-Si-20Cu-Based Filler Metals

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The bond strength of the 6061-T6 aluminum alloy brazed with Al-12Si, Al-9.6Si-20Cu, and Al-7Si-20Cu-2Sn filler metals at a low temperature of 550 °C is evaluated. The fractography of these brazements after tensile tests was observed using scanning electron microscopy (SEM). It was found that joints with good integrity can be produced with Al-7Si-20Cu-2Sn filler metal because it can be used in a temperature range of 504 to 526 °C, about 70 °C lower than the traditional Al-12Si filler metal. It was shown that joints of 6061-T6 aluminum alloy as the base metal, when brazed at 550 °C for 60 min using this new filler metal and ward, and after being subjected to a T6 treatment, possessed a high bonding strength of about 121 MPa.

Keywords

6061-T6 aluminum, Al-12Si, Al-Si-20Cu, brazeability, filler metals

1. Introduction

With advantages, such as high specific strength, low cost, and superior corrosion resistance, aluminum and its alloys have been widely used in the aerospace, automobile, and construction industries. Also, owing to their good thermal conductivity, they are often employed as heat exchangers. Brazing has been considered as one of the most important joining methods for manufacturing aluminum components.[1] Commercially available filler metals for aluminum brazing were usually based on an aluminum-silicon alloy system with compositions around the Al-12Si eutectic. [2,3] However, in order to achieve the strong joints, the very act of brazing with Al-Si filler metals must be performed in the temperature range of 590 to 610 °C, which is near or even above the melting point of many aluminum alloys. As a result, the aluminum workpieces after brazing would be molten either wholly or partially, or their mechanical properties would degenerate substantially.

There clearly exists a demand to develop newer filler metals with lower melting points for brazing most of the aluminum engineering alloys. Many efforts have been made for this purpose in the past ten years. Humpston *et al.*^[4] and Jacobson *et al.*^[5] reported on an Al-5Si-20Cu-2Ni filler metal with a melting range within 518 to 538 °C. When this filler metal is brazed with a 3001 Al alloy, shear strength over 75 MPa is obtained. Earlier, Suzuki *et al.*^[6] had introduced a eutectic Al-4.2Si-40Zn filler metal with a 535 °C melting point, but the disadvantage of their filler metal lies in the very high vapor pressure of zinc, which could constitute quite an impediment to the vacuum-brazing process. In this study, a new Al-7Si-20Cu-2Sn filler metal is developed for the brazing of aluminum alloys. Its

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brazeability with the 6061-T6 aluminum alloy is better than Al-12Si and the Sn-free Al-9.6Si-20Cu filler metals.

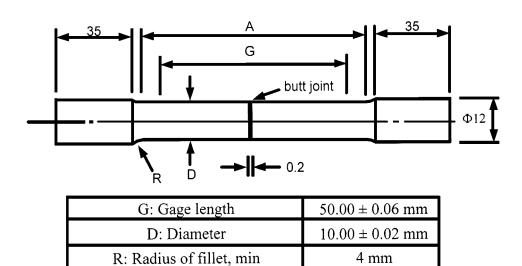
2. Experimental

The Al-Si-20Cu-based filler metals were prepared by melting components in Al₂O₃ crucibles placed in an air furnace. First, the Al-12Si alloy was melted at 700 °C, and then, the temperature was raised to 1000 °C. Copper in a shape of slugs (99.9 wt.% and 2 mm diameter) was added into the molten Al-12Si alloy in the amount of about 20% of the total weight. Afterward, the mixture was stirred for 30 min for homogenization. A part of the Al-Si-20Cu alloy was alloyed with tin. Afterward, the final Al-7Si-20Cu-2Sn composition was cast into a stainless steel mold. In previous authors' article,^[7] the melting ranges for both Al-Si-20Cu-based filler metals, *i.e.*, Al-9.6Si-Cu and Al-7Si-20Cu-2Sn, as analyzed by differential thermal analysis, were 524 to 543 °C and 504 to 526 °C, respectively.

For the evaluation of brazeability of both these filler metals, 6061-T6 aluminum cylindrical specimens having 50 mm length and 22 mm diameter were employed. In preparation for bonding, the specimen surfaces were ground with 400 grit SiC paper; then, they were cleaned by rinsing in distilled water followed by ultrasonic cleaning in acetone for 5 min. Table 1 shows the chemical compositions, solidus and liquidus, and the ultimate tensile strength of the 6061-T6 joints.

A 0.2 mm thick filler-metal foil was inserted between two 6061-T6 aluminum bars. The new alloy was brazed at 550 °C in a vacuum furnace under a pressure of 5×10^{-5} torr, whereas a traditional Al-12Si filler metal was brazed at 600 °C. After brazing, some specimens were postheat treated under the T6-temper condition, *i.e.*, the solid solution at 520 °C for 1.5 h, water quenching, and then aging at 175 °C for 8 h. The microstructures of the brazed joints were observed using optical microscopy and scanning electron microscopy (SEM). Etching in 10 vol.% H_3PO_4 at 50 °C for 60 s was employed.

In order to evaluate the bond strength, the average ultimate tensile stress at room temperature was determined at a stain rate of 10^{-3} s⁻¹ averaging data on three brazed specimens. Figure 1 shows dimensions of the tensile test specimens.



60 mm

A: Length of reduced section, min

Fig. 1 Dimensions of the joint tensile specimen

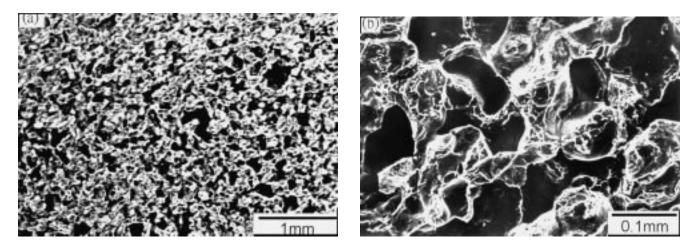


Fig. 2 (a) The SEM tensile fractured surface of the Al-12Si/6061-T6 joint after brazing at 600 °C for 30 min and (b) an enlarged detail of (a)

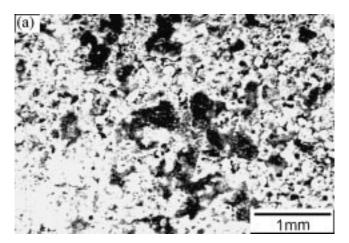
Table 1 Chemical composition, solidus (T_S) and liquidus (T_L) , and the ultimate tensile strength (σ_{UTS}) of the 6061-T6 alloy

		C	Chemical com	position (wt.%	(6)				
Alloy	Mn	Mg	Si	Cu	Cr	Al	T_S (°C)	T_L (°C)	(σ_{UTS}) (MPa)
6061-T6	0.01	1.10	0.61	0.25	0.12	Bal.	592	654	331

3. Results and Discussion

The traditional Al-12Si filler metal possesses a eutectic point at about 577 °C; therefore, the brazing process at 550 °C cannot be applied. Not until the brazing temperature is raised to 600 °C can the 6061-T6 aluminum alloy be bonded with this filler metal. However, the joint strength remained at about 60 MPa. Evidently, brazing of a 6061-T6 alloy under such conditions results in poor quality joints. Figure 2 shows

the tensile fractured surface of Al-12Si/6061-T6 brazements brazed at 600 °C for 30 min, which contains numerous voids seen as dark areas on the photograph. Since the brazing temperature was only slightly higher than the liquidus temperature of the Al-12Si filler metal, the wettability of the molten Al-12Si filler metal with respect to the 6061-T6 alloy was poor, causing formation of voids in the joining zone and a low joint strength. However, a further increase of the brazing temperature above 600 °C is inapplicable or impractical, because this



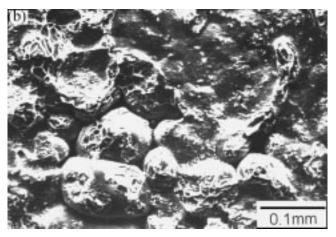


Fig. 3 (a) The SEM tensile fractured surface of the Al-9.6Si- 20Cu/6061-T6 joint after brazing at 550 °C for 30 min and (b) an enlarged detail of (a)

Table 2 Brazing conditions and the joint strength of the 6061-T6 aluminum brazements

Filler metal	Brazing conditions	Postheat treatment after brazing(a)	Joint strength, MPa
Al-12Si	550 °C, 30 min	•••	Failed
Al-12Si	550 °C, 60 min		Failed
Al-9.6Si-20Cu	550 °C, 30 min		41 ± 15
Al-9.6Si-20Cu	550 °C, 60 min	T6	56 ± 11
Al-7Si-20Cu-2Sn	550 °C, 30 min		62 ± 11
Al-7Si-20Cu-2Sn	550 °C, 60 min	T6	121 ± 9

(a) The T6-temper condition: solution treated at 520 $^{\circ}C$ for 1.5 h, water quenched, and then aged at 175 $^{\circ}C$ for 8 h

might incur the risk of partial melting of the 6061 aluminum workpieces in an industrial furnace.

Figure 3 shows the fractography of the 6061-T6 aluminum joint after brazing at 550 °C for 30 min with an Al-9.6Si-20Cu filler metal. Since the brazing temperature was only 7 °C higher than the liquidus (543 °C) of this filler metal, certain areas of the surface were left unbonded and covered with solidified grains of the filler metal, as shown in Fig. 3. In this case, the joint strength, as shown in Table 2, was quite low as well. As the brazing time increased from 30 to 60 min and a T6-postheat treatment was rendered, the bonding strength could improve from 41 to 56 MPa. Fractography showed that there were still certain areas of the surface left unbonded. However, a few regions characterized with dimple fractures could be observed in this case, which implies that good joining occurred in these local regions (Fig. 4). A satisfactory joint could be obtained only when the brazing temperature was raised to about 570 °C in the case of the Al-9.6Si-20Cu filler metal. When an Al-7Si-20Cu-2Sn filler metal was used to join the 6061-T6 aluminum alloy at 550 °C for 30 min, the bond strength was increased to about 62 MPa. Figure 5 shows that the areas with the dimplefractured regions increased slightly in the case of Al-7Si-20Cu-2Sn. However, when the brazing time was extended to 60 min and then a T6 treatment rendered, the joint strength rose to 121

MPa, a much higher value than what was obtained by Jacobson *et al.* when they used the Al-5Si-20Cu-2Ni filler metal to braze a 3001 Al alloy.^[5] In comparison with the fractography of the cases showed in Fig. 2 through 5, many more dimple-fractured regions appeared on the surface of these specimens (Fig. 6). It was also found that the Al-7Si-20Cu-2Sn filler metal has a good wettability with respect to the 6061-T6 alloy.

It was shown that the Al-7Si-20Cu-2Sn filler metal produced a sound joint when brazed with the 6061-T6 aluminum alloy and carried out at a low temperature of 550 °C. Figure 7 shows three different microstructures of the 6061-T6 joints brazed with Al-12Si, Al-9.6Si-20Cu, and Al-7Si-20Cu-2Sn filler metals. As is evident from these micrographs, in the case of the Al-12Si brazement prepared at 600 °C for 30 min, the brazing zone is about 600 μ m. However, where Al-9.6Si-20Cu and Al-7Si-20Cu-2Sn filler metals were used and brazing was conducted at the lower temperature of 550 °C, the brazing zone is narrower, only 400 μ m, thus resulting in a substantial improvement of the joint strength.

During brazing, the molten filler metal reacted with the tested aluminum-base metals resulting in a decrease of the concentration of elements depressing the melting temperature, such as Si, Cu, and Sn. When the solidification stage sets in, the α -Al solid-solution grains nucleate at the liquid/solid interface and grow toward the joint center. Since these elements have lower solubility in the α -Al solid solution, they were rejected into the intergranular residual-liquid region. Finally, the residual-liquid phase when solidified would lie out according to its intergranular characteristics, as the morphology of the joint in Fig. 7 would illustrate. The sound joint formed in the Al-7Si-20Cu-2Sn/6061-T6 brazements is attributed to the much lower liquidus of the Al-7Si-20Cu-2Sn filler metal in comparison with the brazing temperature, which explains its good wettability and flowability in the clearance of the studied specimens.

4. Conclusions

The addition of 20 wt.% Cu into the Al-12Si alloy decreases the melting range by about 50 °C. Further addition of 2 wt.% Sn into the Al-Si-Cu alloy decreases its melting temperature

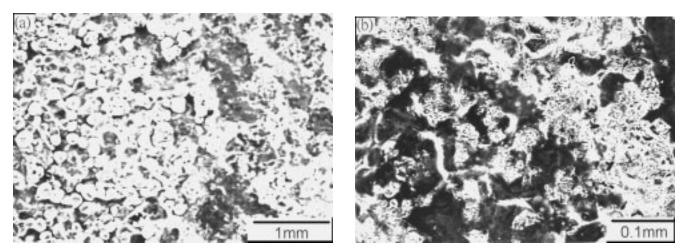


Fig. 4 (a) The SEM tensile fractured surface of the Al-9.6Si-20Cu/6061-T6 joint after brazing at 550 °C for 60 min followed by a T6 post-treatment and (b) an enlarged detail of (a)

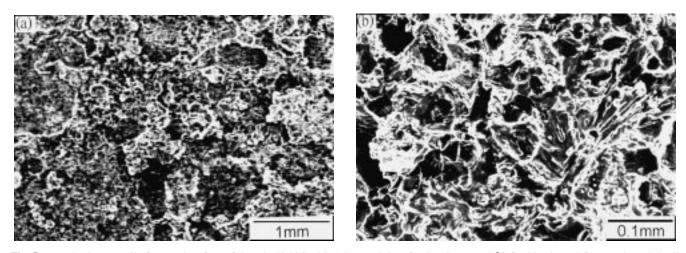


Fig. 5 (a) The SEM tensile fractured surface of the Al-7Si-20Cu-2Sn/6061-T6 joint after brazing at $550\,^{\circ}$ C for 30 min and (b) an enlarged detail of (a)

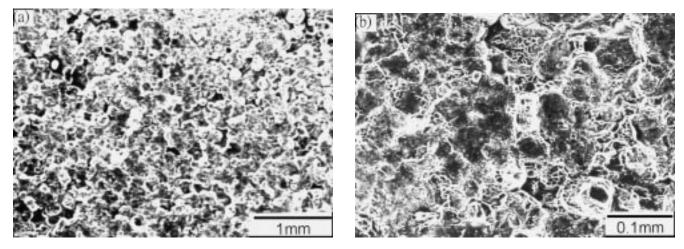


Fig. 6 (a) The SEM tensile fractured surface of the Al-7Si-20Cu-2Sn/6061-T6 joint after brazing at 550 °C for 60 min followed by a T6 posttreatment and (b) an enlarged detail of (a)

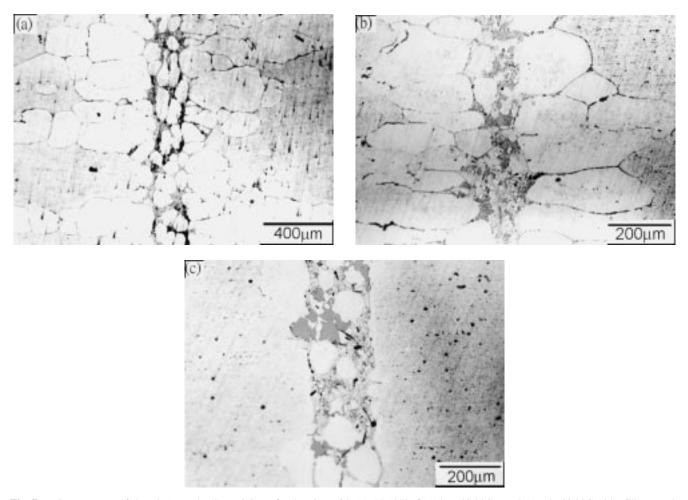


Fig. 7 Microstructure of the 6061-T6 aluminum joints after brazing with (a) Al-12Si, (b) Al-9.6Si-20Cu, and (c) Al-7Si-20Cu-2Sn filler metals

range even farther, 504 to 526 °C. As the brazeability of the 6061-T6 aluminum alloy with all these filler metals at the low temperature of 550 °C is evaluated, this study concludes that the Al-7Si-20Cu-2Sn filler metal possesses an excellent wettability and a relatively high bonding strength of about 121 MPa.

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